

# Materials and Manufacturing Challenges for Compact Heat Exchangers for Supercritical CO<sub>2</sub> Power Systems

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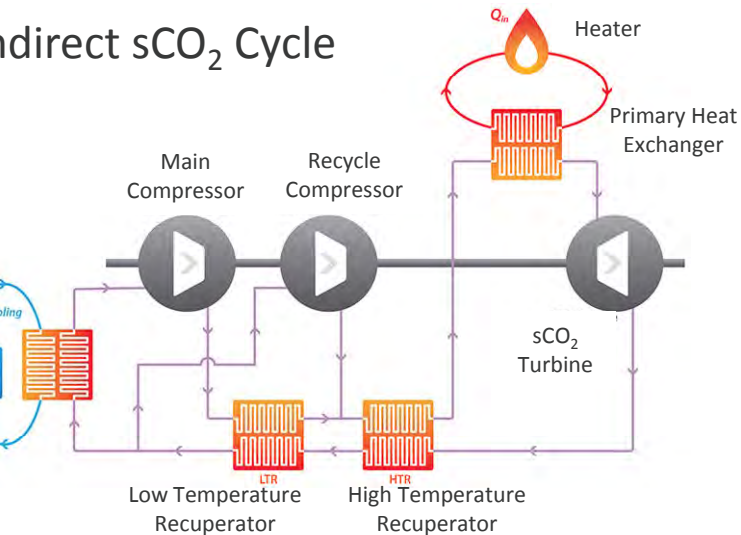
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**DISCLAIMER**

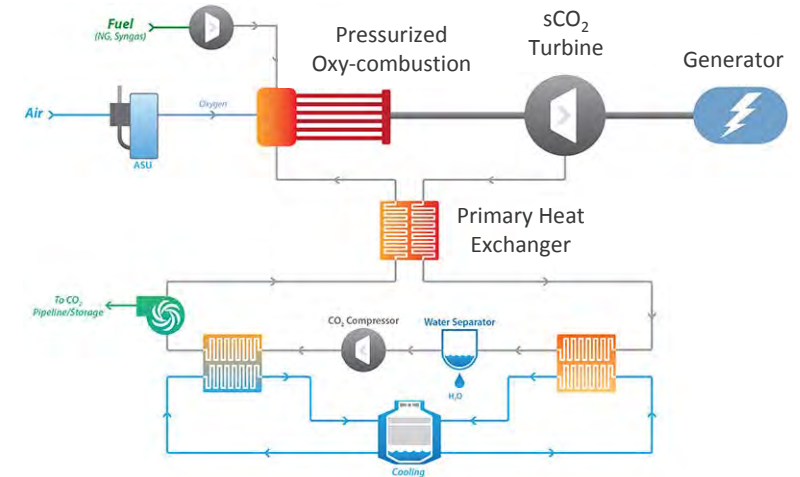
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# Supercritical CO<sub>2</sub> Power Cycles

Indirect sCO<sub>2</sub> Cycle



Direct sCO<sub>2</sub> Cycle



Cycle/Component	Inlet		Outlet	
	T (C)	P (MPa)	T (C)	P (MPa)
Indirect	Heater	450-535	650-750	1-10
	Turbine	650-750	550-650	8-10
	HX	550-650	100-200	8-10
Direct	Combustor	750	1150	20-30
	Turbine	1150	800	3-8
	HX	800	100	3-8

Essentially pure CO<sub>2</sub>

CO<sub>2</sub> with combustion products including O<sub>2</sub>, H<sub>2</sub>O and SO<sub>2</sub>

# Compact Heat Exchangers

## Higher efficiency

- Due to much shorter heat diffusion lengths in fluid

## Smaller size

- Use of less materials (expensive superalloys)
- Takes less space

## Modular design

- Expandable to large power plants

# Typical Compact HX Fabrication Process

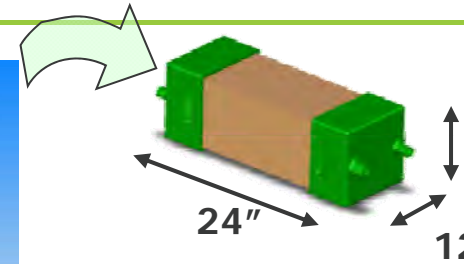
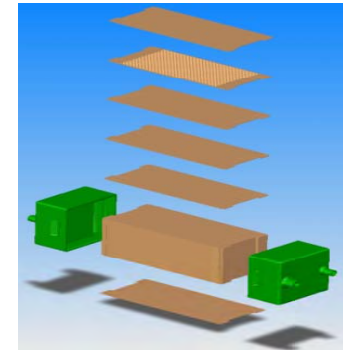
Pattern microscale flow paths into laminae using a variety of methods

- Chemical etching
- Micromachining
- Laser cutting
- EDM

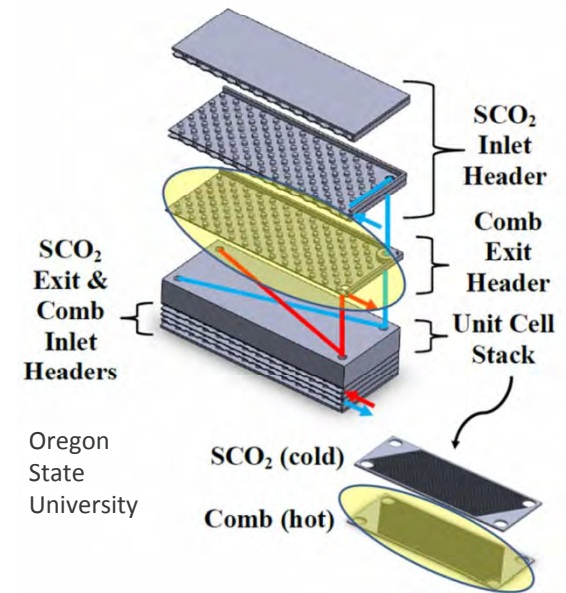
Bond these laminae using a variety of methods

- Diffusion bonding
- Transient liquid phase (TLP) bonding
- Laser welding
- Brazing

For sCO<sub>2</sub> cycles, diffusion bonding and TLP bonding considered to be the most robust approaches



\*W. Ehrfeld, V. Hessel, H. Löwe, **Microreactors: New Technology for Modern Chemistry**, Wiley-VCH, 2000.



# Bonding

Bonding is considered the “weak link” in the fabrication process

Sharp edges in the architecture lead to locations of high stress concentration in the mechanical design simulations

We need information on

- The parameters for bonding process (T, P, t) of materials
- The strength of the bond
- Corrosion behavior of bonded regions in sCO<sub>2</sub>



# Materials

High-temperature strength  
High-temperature oxidation resistance



## Nominal chemical composition (weight %) of materials used in this study (Haynes 230 and Haynes 282)

	Ni	Cr	W	Ti	Mo	Fe	Co	Mn	Si	Al	C	B
	57	22	14	--	2	3*	5*	0.5	0.4	0.3	0.10	0.015*
	57	19.5	--	2.1	8.5	1.5*	10	0.3*	0.15*	1.5	0.06	0.005

\* = maximum

*Other materials considered for this application*

Inconel 740H

Inconel 625

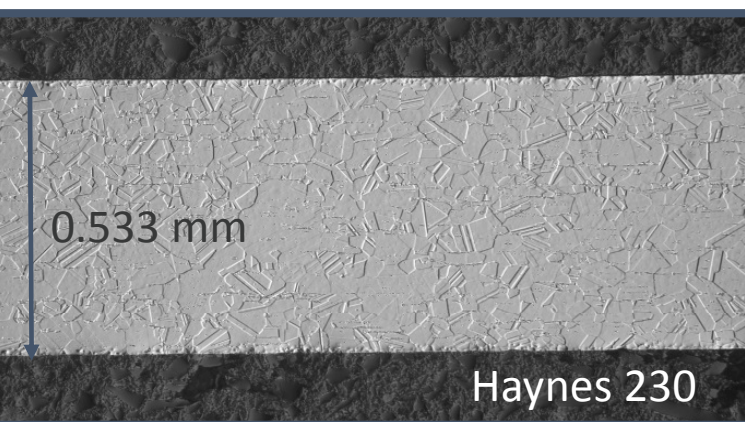
Inconel 617

347H Stainless steel

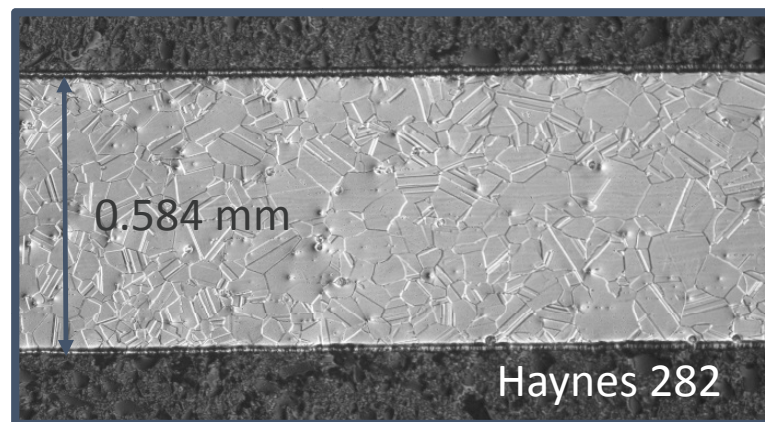
316 Stainless steel

304 Stainless steel

Grade 91 steel



Haynes 230



Haynes 282

Solid-solution strengthened

Cold rolled and 1232 °C solution annealed sheet

Precipitation strengthened

1149 °C solution annealed sheet

# Diffusion Bonding vs. Transient Liquid Phase (TLP) Bonding

## Diffusion Bonding

Diffusion bonding is a solid state process

It requires applied high pressure at high temperature for a certain amount of time

It involves diffusion of constituent atoms and creep processes to close the voids present due to roughness of the faying surfaces.

## Transient Liquid Phase Bonding

- TLP involves both solid state and liquid state reactions
- It requires less pressure than diffusion bonding
- It requires a lower melting point interlayer
- It involves isothermal melting and solidification of interlayer



# Bonding

Sheets were water-jet cut into shims

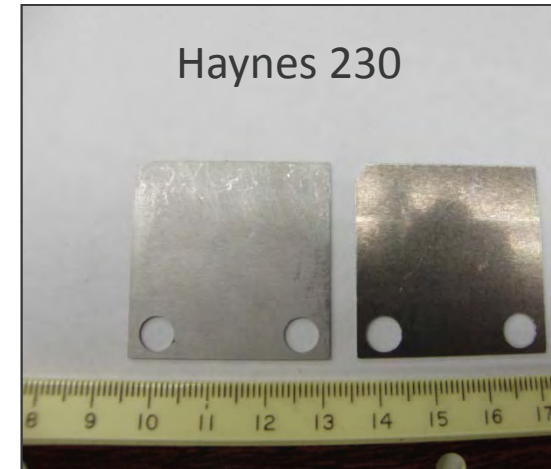
100 shims were bonded together in each stack

All shims were reverse current etched and cleaned with acetone

Some stacks used shims plated with electroless nickel, 2 - 4  $\mu\text{m}$  thick

Some shims contained pin-fin micro-features identical to those used in a heat exchanger

All shims were thoroughly cleaned by hand and in an ultrasonic acetone bath for 15 minutes immediately before bonding



# Bonding

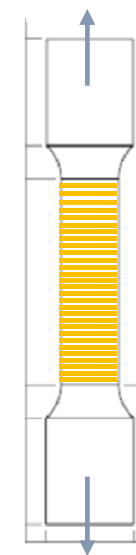
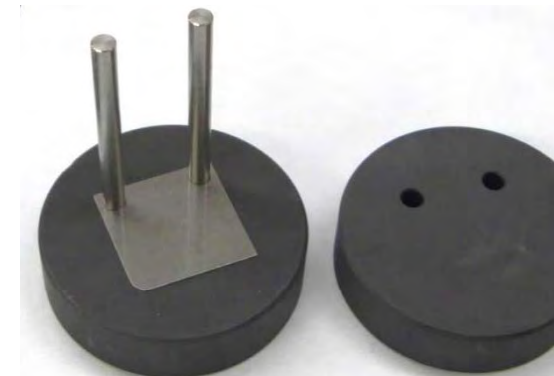
Thin stacks were held in a fixture during bonding and pressure was applied only after the temperature ramped up to the desired value

The hot press vacuum was maintained at approximately  $5 \times 10^{-6}$  torr (0.0007 Pa)

150°C for 8 hours at 12.7 MPa

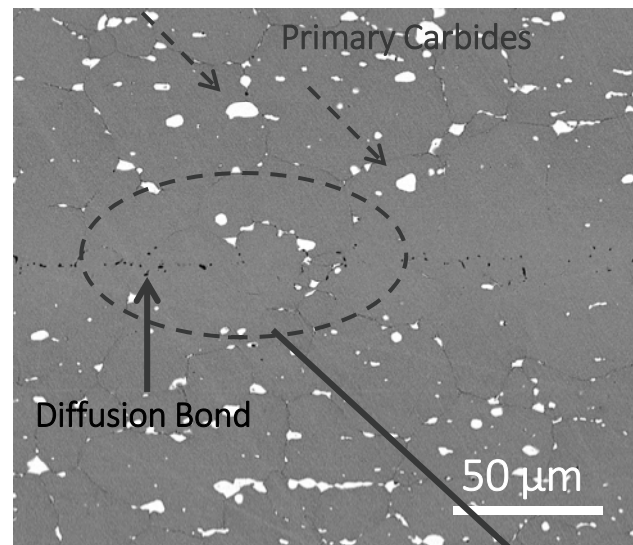
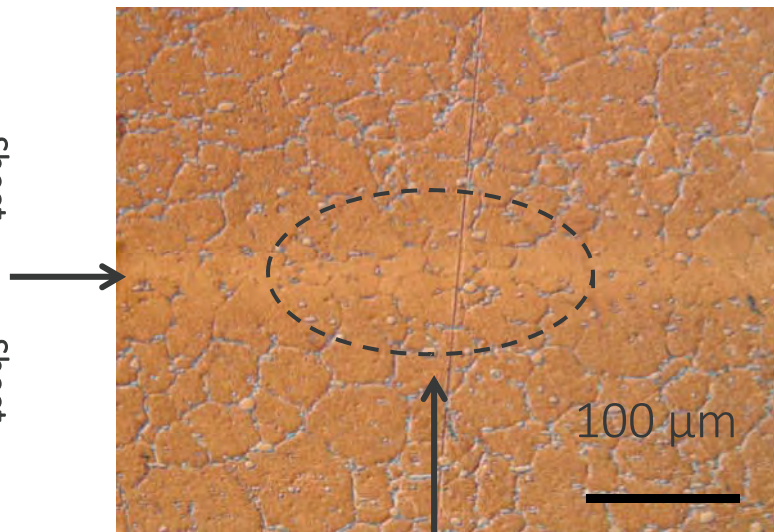
After bonding, each stack was machined to produce 6 tensile specimens using wire EDM and CNC lathe

H282 without Ni plating did not bond well

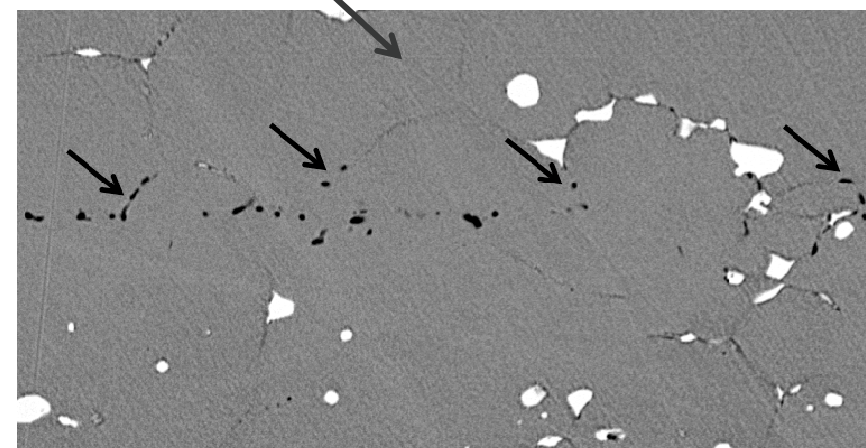


# Diffusion Bonding of Alloy 230

## Microstructure



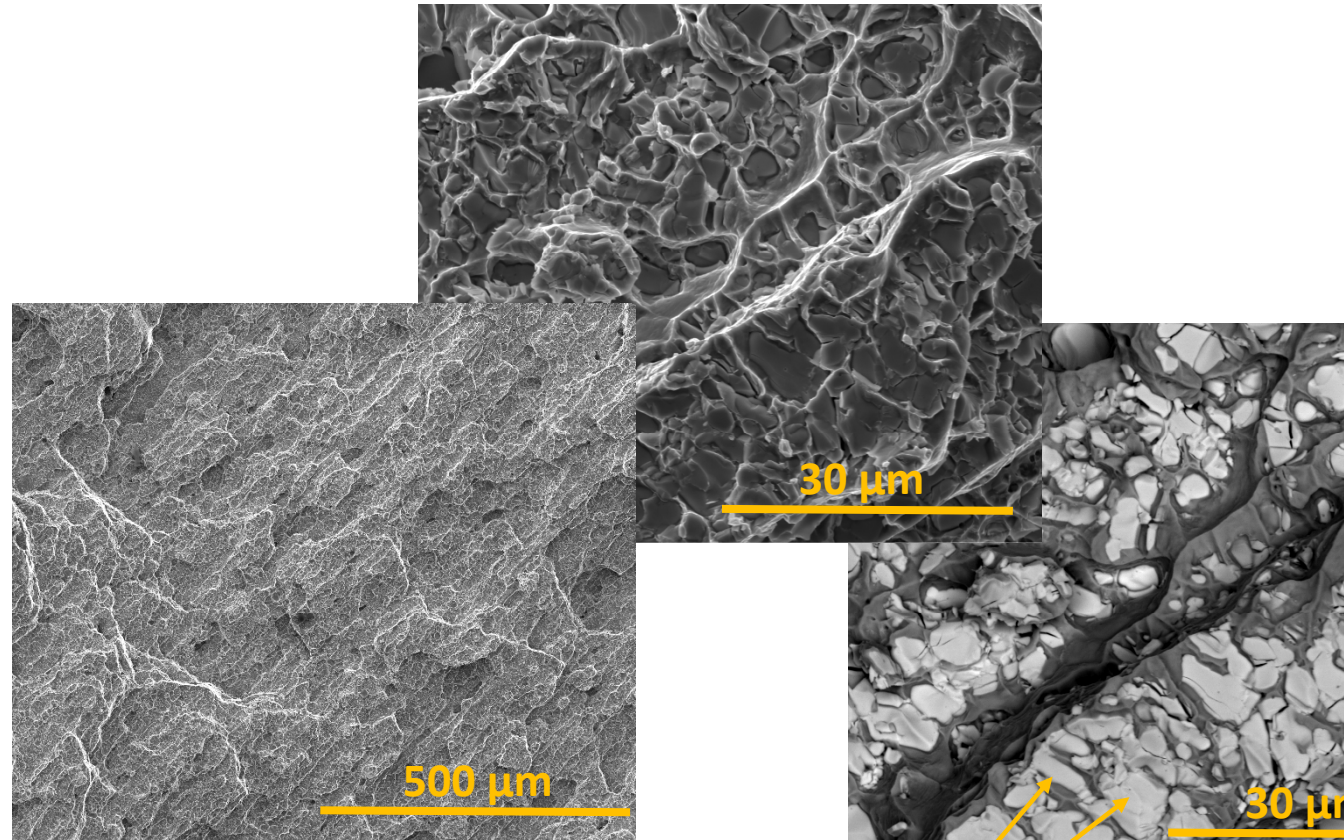
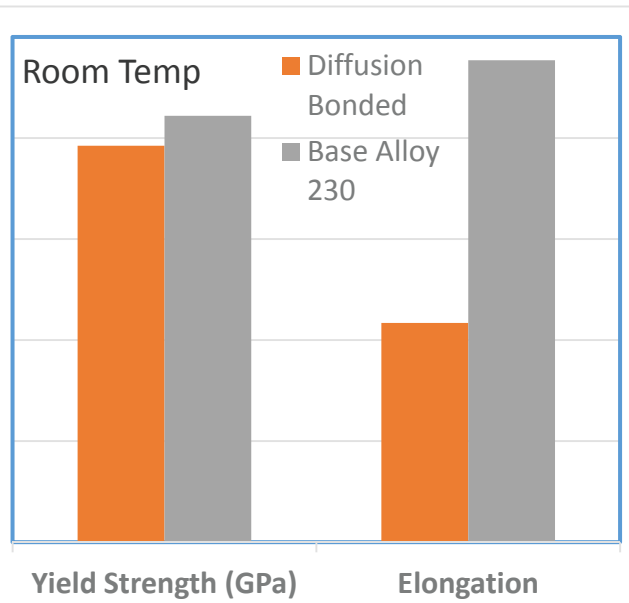
- Primary carbides form at higher temperature



atched microstructure to observe grain growth through the bond line

# Diffusion Bonding of Alloy 230

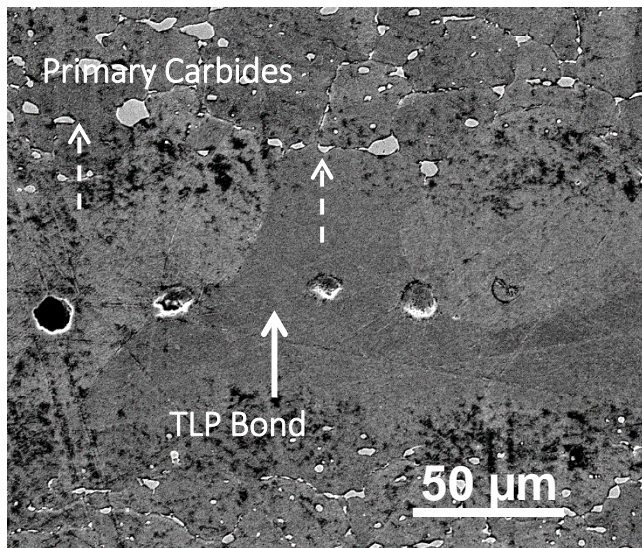
## Mechanical Behavior



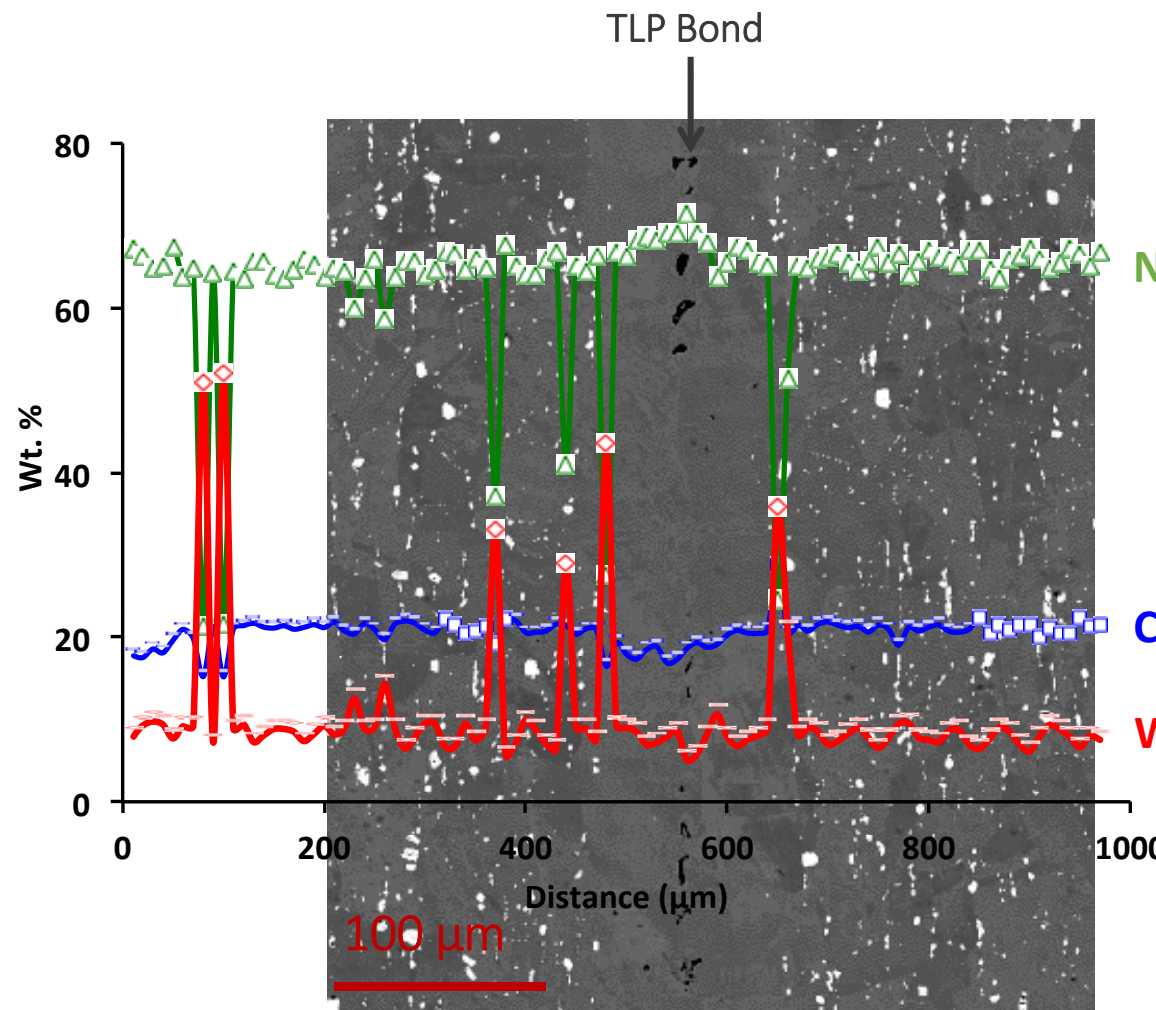
Fracture occurs in a ductile manner along the precipitate bands.

# P Bonding of Alloy 230

## Microstructure

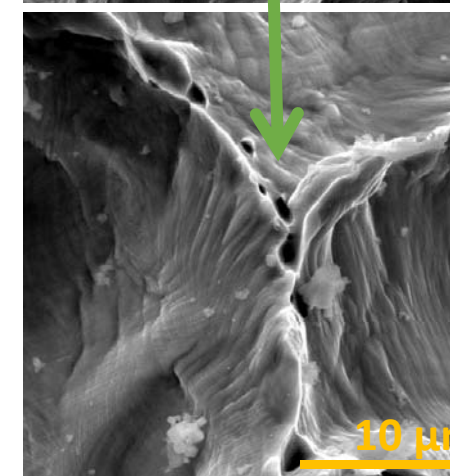
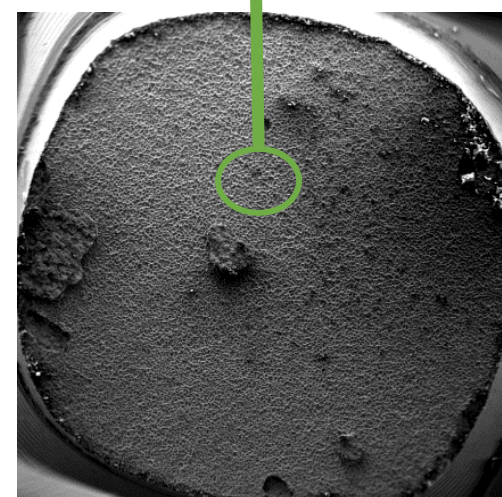
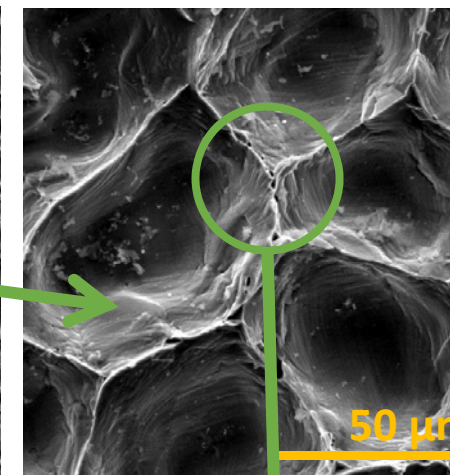
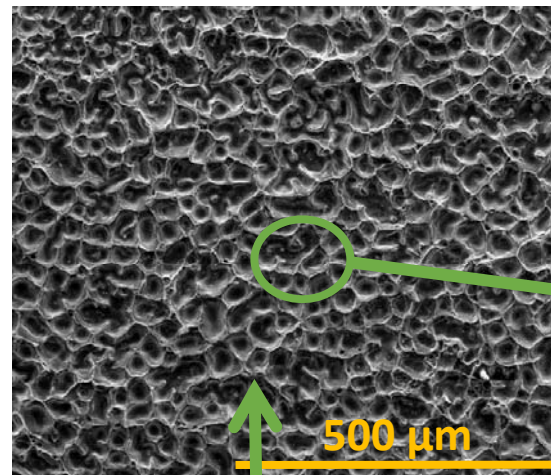
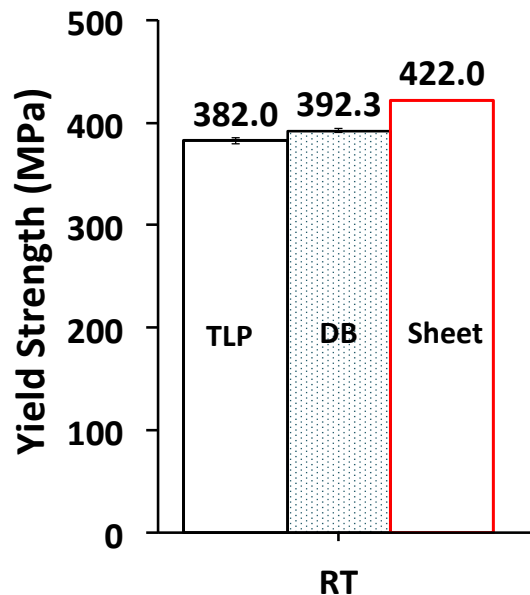
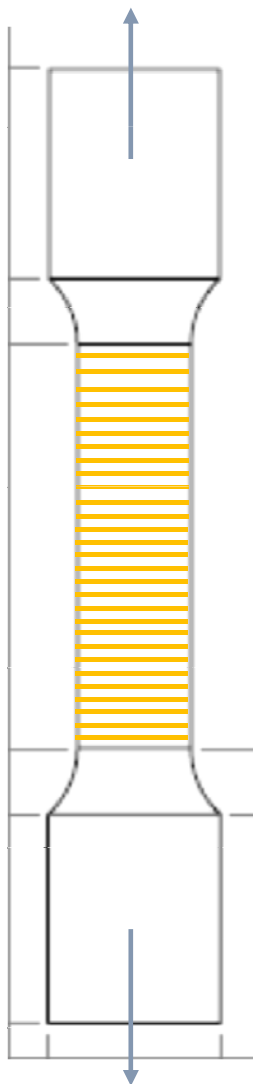


- Primary Carbides
- Increase in Ni, dip in Cr at the bond



# Bond Strength – TLP Bonding of Alloy 230

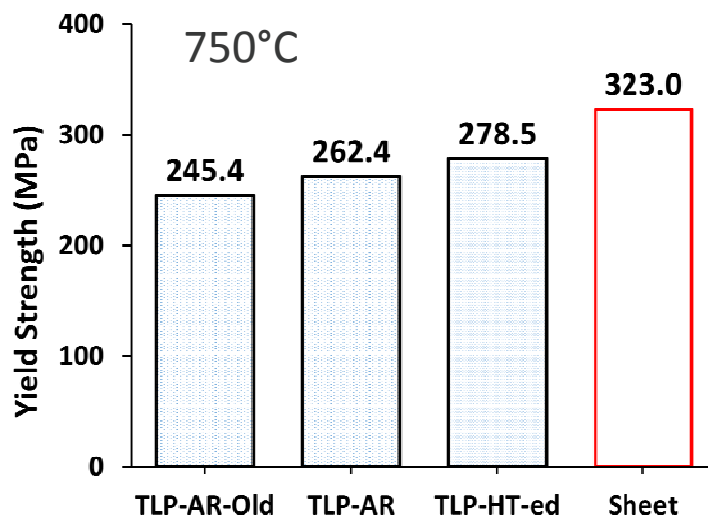
Room Temperature Strength



Fracture occurs in a ductile manner at the bonding layer.

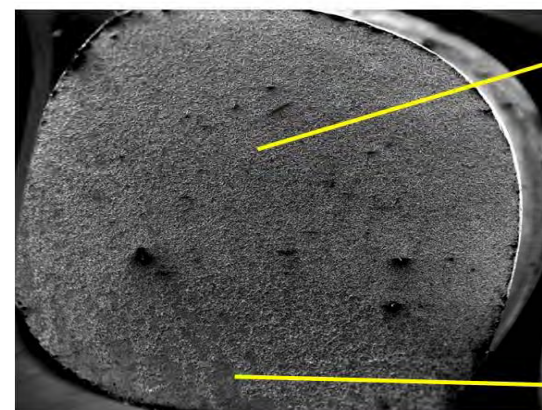
# Bond Strength – TLP Bonding of Alloy 230

## High-Temperature Strength

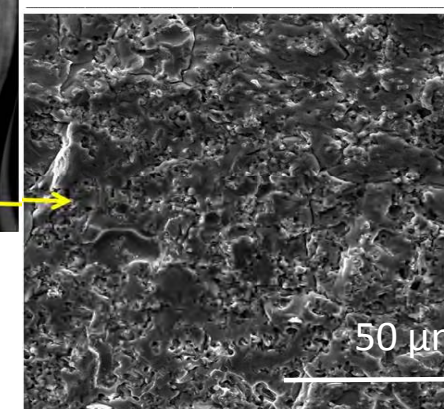
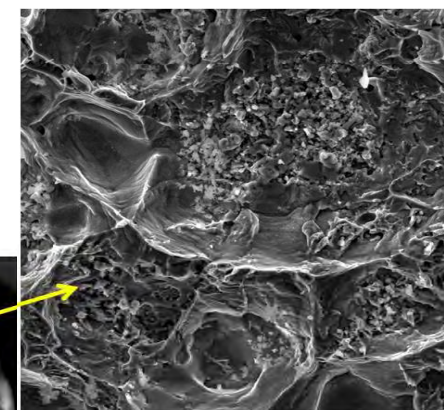


### High Temperature Yield Strength

- TLP-AR-Old = 76%
- TLP-AR = 81%
- TLP-HT-ed = 86%

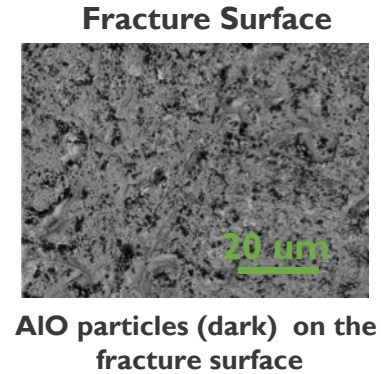
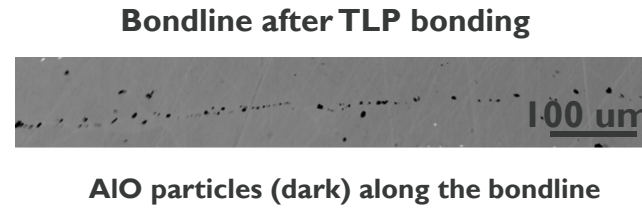
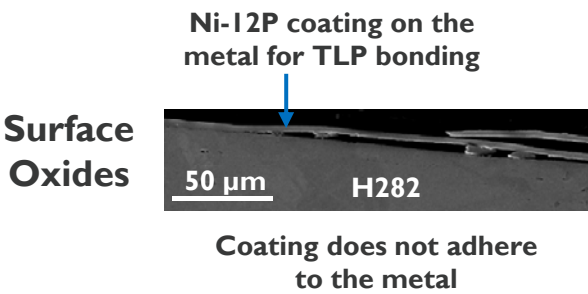


Fracture occurs in a ductile manner at the bonding layer and the sheet.

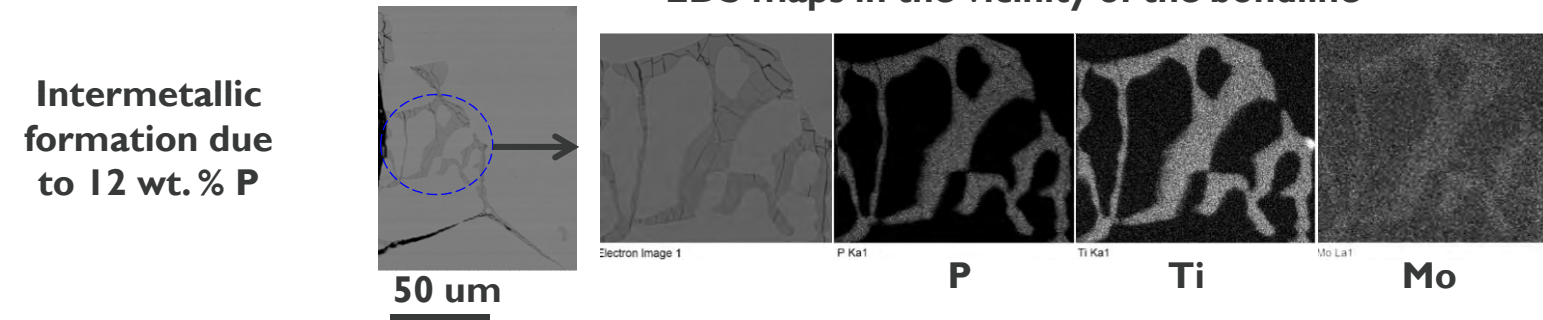


# Bonding Defects

## Challenges with TLP bonding of H282 – Surface oxides & Intermetallic formation

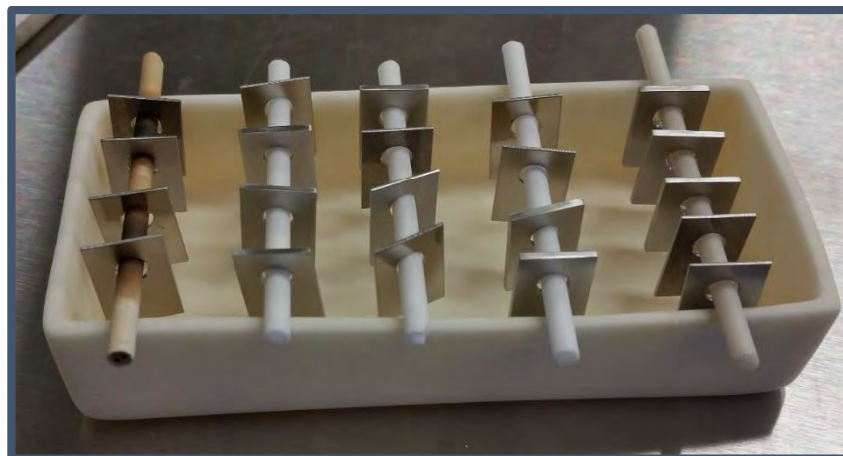
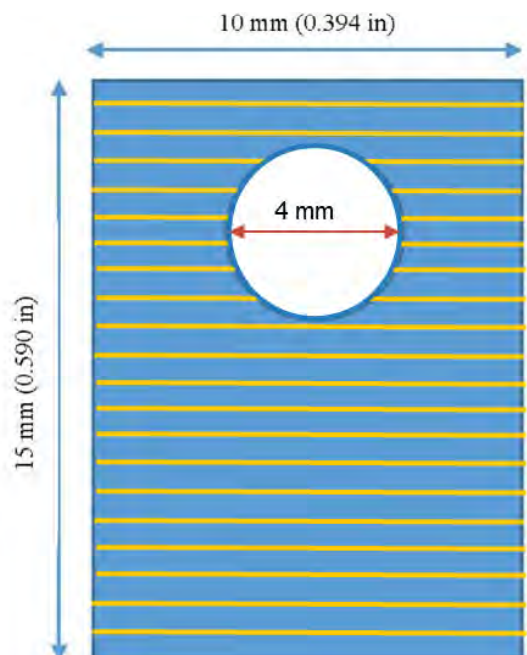


### EDS maps in the vicinity of the bondline





# Oxidation of Bonded Regions



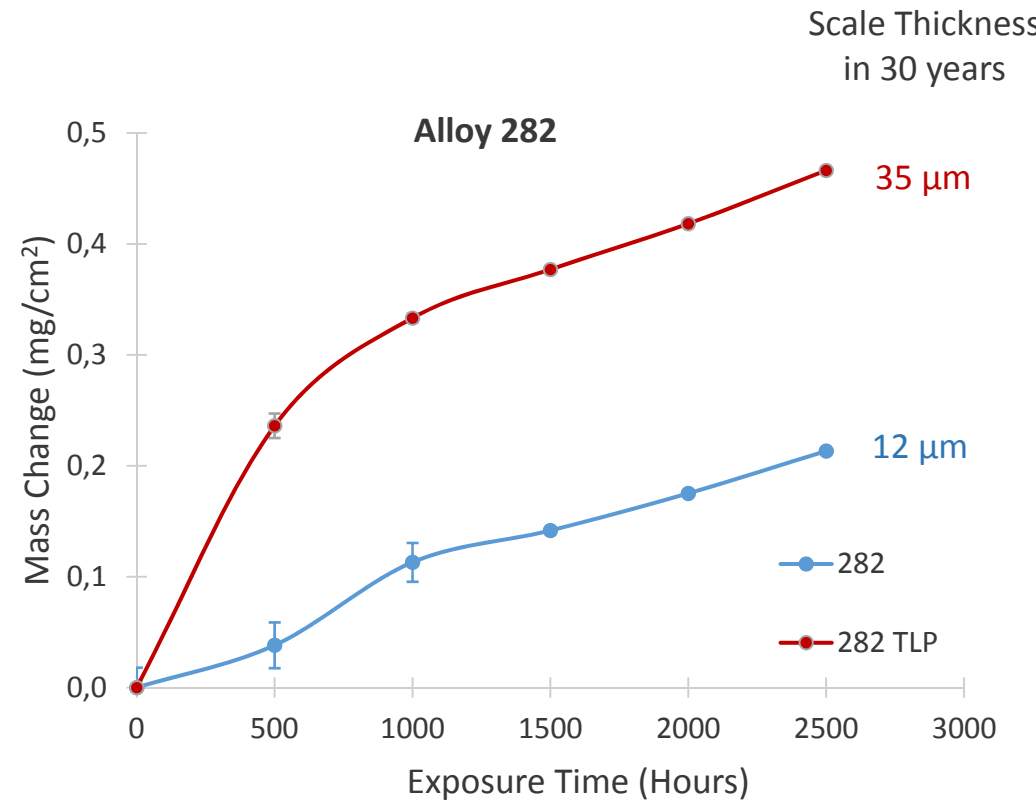
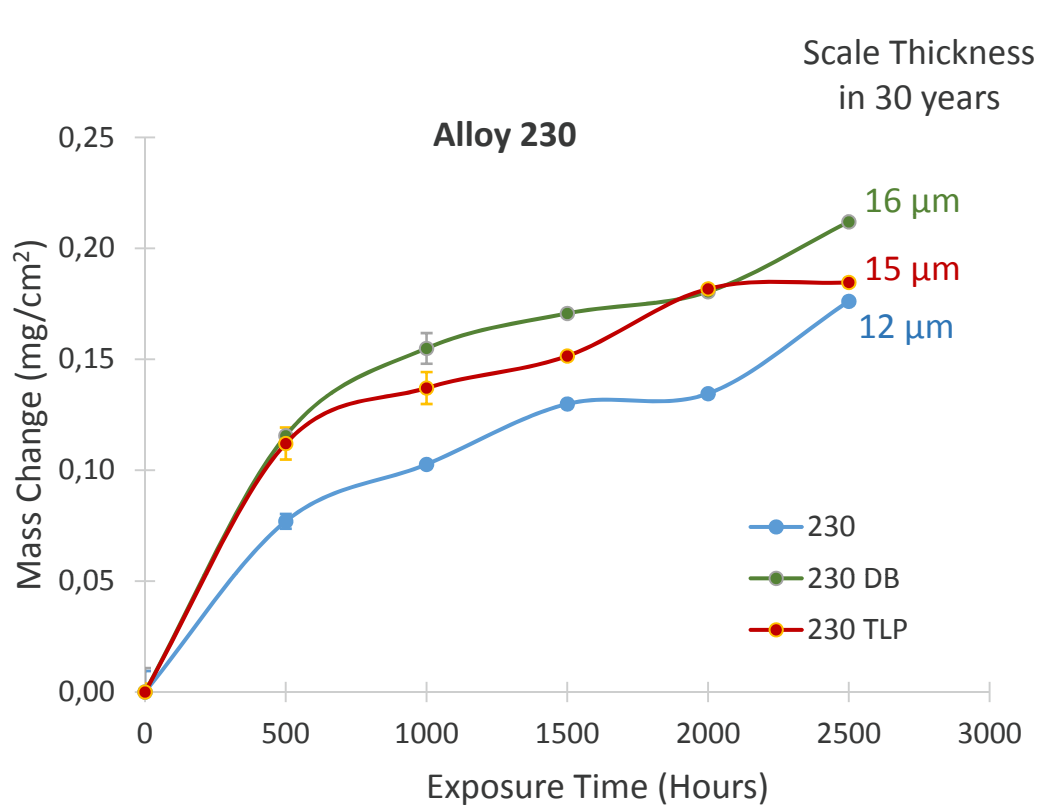
Characterization  
Mass Change  
XRD  
SEM

Gas: 1 bar CO<sub>2</sub> (99.999% purity)  
Gas flow rate: 0.032 kg/h  
Temperature: 700°C  
Duration: 2500 h  
24 h purging with CO<sub>2</sub> before heating



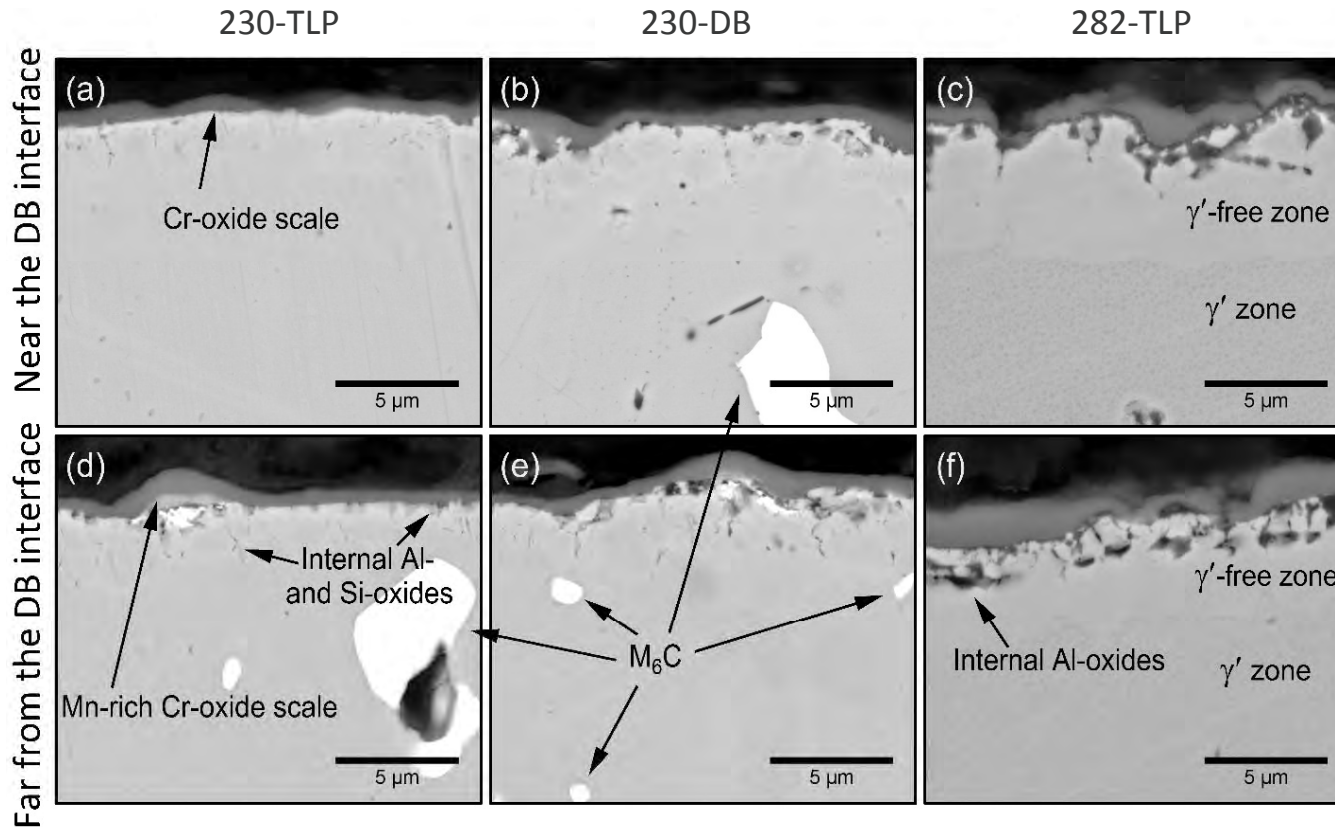
# Oxidation of Bonded Regions

1 bar CO<sub>2</sub> at 700°C



# Oxidation of Bonded Regions

1 bar CO<sub>2</sub> at 700°C



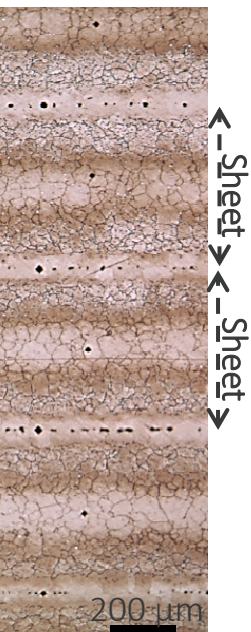
No significant difference between bond regions and away from bond regions

More internal oxidation in H282, resulting from higher Al and Ti levels

γ' loss in H282 below the internal oxidation layer

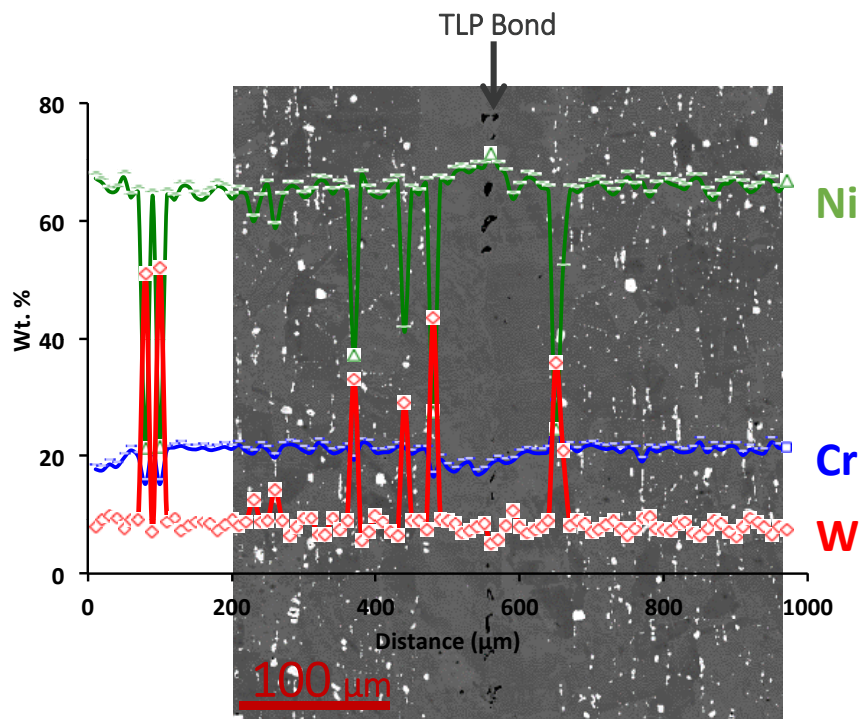
Back-Scattered Electron Images

# Summary

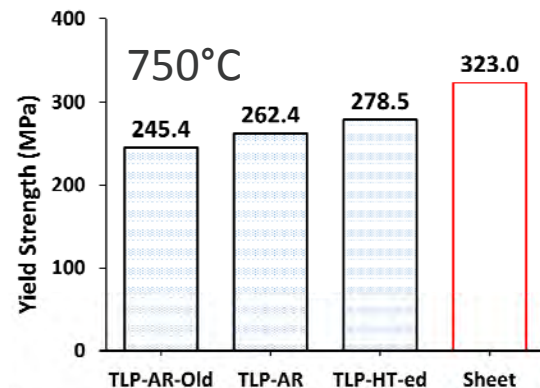


voids at bondline

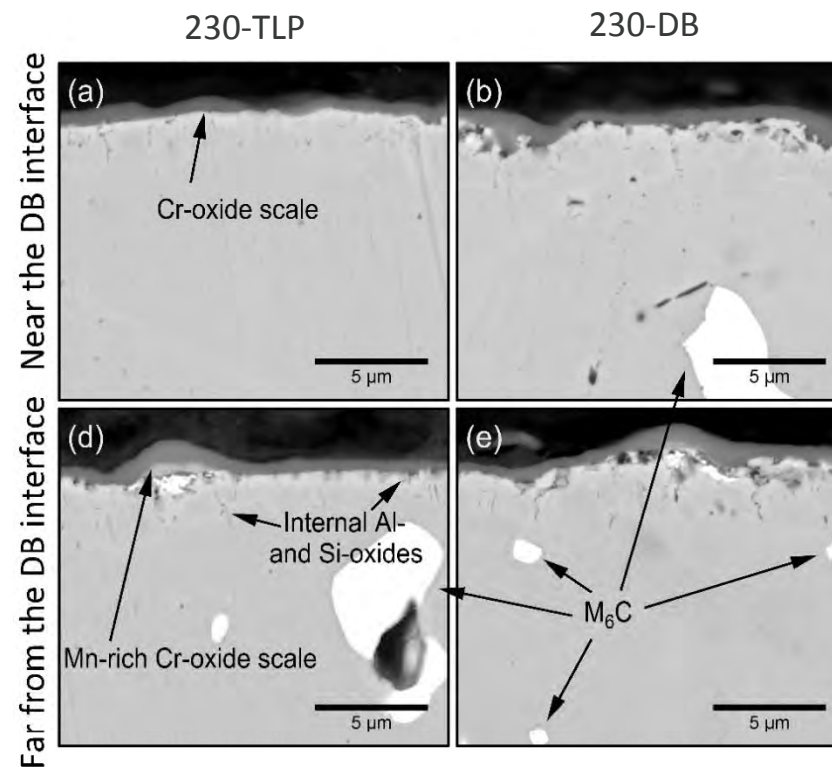
Intermetallic formation



Composition of TLP bonded regions were similar to the base material



Both DB and TLP bonded stacks exhibited good strength at high and room temperature



No significant difference in oxidation behavior in CO<sub>2</sub> at 700°C between bonded regions and away from bond region